

# Effect of Spatial Heterogeneity on the Unusual Uptake Behavior of Multivariate-Metal–Organic Frameworks

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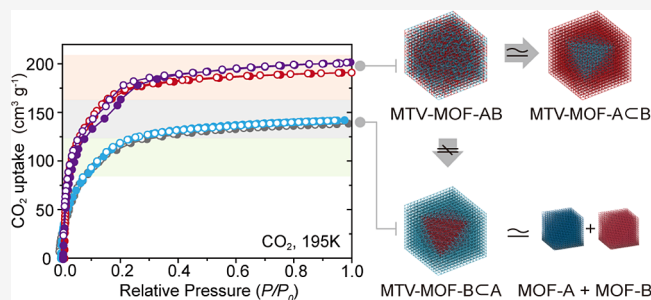


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**ABSTRACT:** The uniqueness of multivariate metal–organic frameworks (MTV-MOFs) has been widely explored to discover their unknown opportunities. While mesoscopic apportionments have been studied, macroscopic heterogeneity and its spatial effects remain unexplored in MTV-MOFs. In this study, we investigated the effect of macroscopic heterogeneity on MTV-MOFs on their uptake behaviors by comparing three types of MTV-MOFs having the functional groups in inner, outer, or entire parts of crystals. Their adsorption behavior for carbon dioxide (CO<sub>2</sub>) and water (H<sub>2</sub>O) brought out that functional groups located in the outer part of the crystals dominantly influence the sorption behavior of MTV-MOFs. These results are also visualized by observing iodine adsorption in the three types of MTV-MOFs using scanning transmission electron microscopy–electron energy loss spectroscopy. We believe that this finding provides new ways to decipher and design MTV-MOFs for their unusual properties.

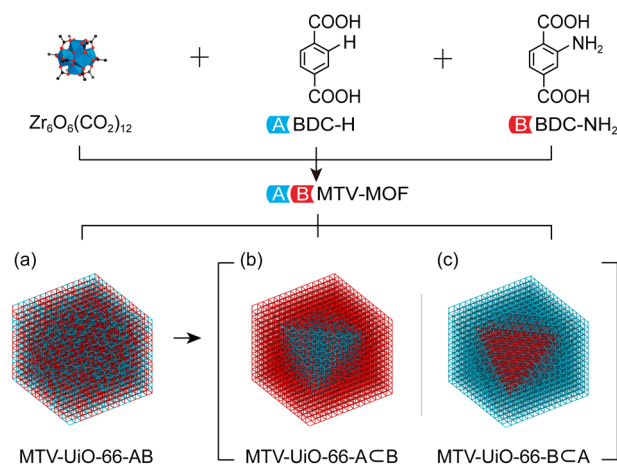


## INTRODUCTION

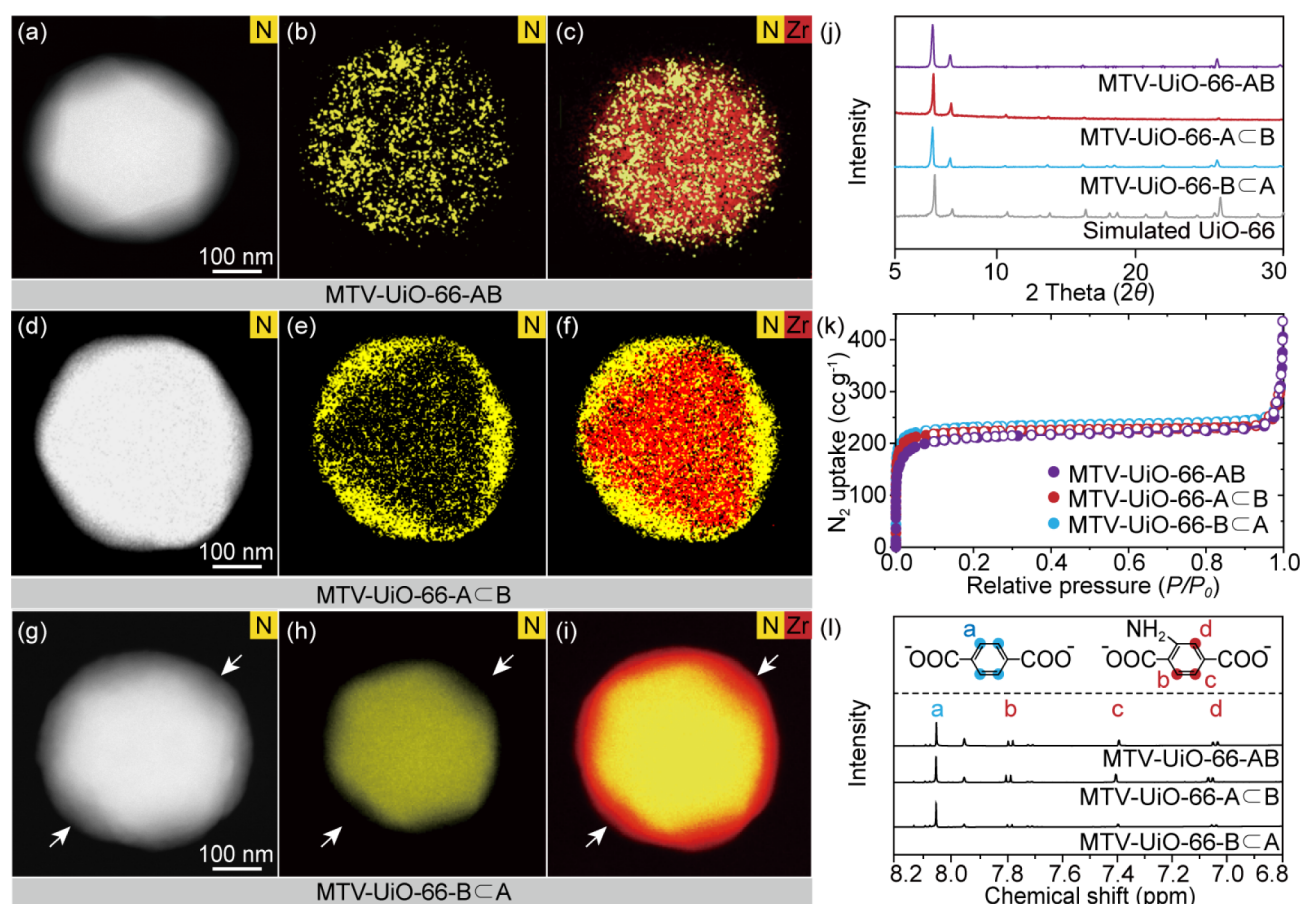
Multivariate metal–organic frameworks (MTV-MOFs) introduce heterogeneity by incorporating multiple functional groups without changing the material's underlying topology. This heterogeneity leads to unusual adsorption properties that have not been observed in homogeneous MOFs having single functional groups.<sup>1–4</sup> The unique adsorption properties in MTV-MOFs have been deciphered by apportionments of multiple functional groups existing all over the crystals,<sup>5–8</sup> which is one of the key factors to discover unknown opportunities of MTV-MOFs. It was observed that different mesoscopic apportionment scenarios such as small cluster formation, random distribution, and well-mixed organic struts have an important impact on the sorption properties of MTV-MOFs.<sup>5</sup> Expanding the scope of this strategy, we hypothesized that controlling the spatial locality of the apportionment would influence the unique sorption behavior and unveil important factors contributing to MTV-MOFs.

In this study, we chose zirconium-based MOFs themed UiO-66 and two different organic struts considering the affinity differences with respect to polar adsorbates such as CO<sub>2</sub> and H<sub>2</sub>O: 1,4-benzenedicarboxylate (denoted as BDC-H or simply A) and 2-amino-1,4-benzenedicarboxylate (denoted as BDC-NH<sub>2</sub> or simply B). Three types of materials were prepared to elucidate the decisive role of spatial locality in MTV-MOFs; (1) MTV-UiO-66-AB containing BDC-H (A) and BDC-NH<sub>2</sub> (B) linkers all over the crystals, as shown in Scheme 1a, (2) MTV-UiO-66-A C B having a BDC-NH<sub>2</sub> linker outer part and a BDC-H linker inner part of its crystals, as shown in Scheme 1b, and (3) MTV-UiO-66-B C A having a BDC-H linker outer

**Scheme 1. Schematic Diagram for MTV-MOF with Two Distinct Linkers of BDC-H (A) and BDC-NH<sub>2</sub> (B): (a) MTV-UiO-66-AB, (b) MTV-UiO-66-A C B, and (c) MTV-UiO-66-B C A**



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**Figure 1.** Characterization of spatially localized MTV-MOF crystals. (a–i) STEM images (left) and EELS-derived maps of nitrogen (center) and nitrogen/zirconium (right) for MTV-MOFs. (a–c) MTV-UiO-66-AB (d–f) MTV-UiO-66-A ⊂ B (g–i) MTV-UiO-66-B ⊂ A. (j) PXRD patterns, (k) N<sub>2</sub> sorption isotherms at 77 K, and (l) digested <sup>1</sup>H NMR spectra of MTV-MOFs.

part and a BDC-NH<sub>2</sub> linker inner part of its crystals, as shown in Scheme 1c. Furthermore, three types of homogeneous MOFs, UiO-66-A, UiO-66-B, and mechanical mixtures of UiO-66-A and UiO-66-B (mix-UiO-66-AB), were also prepared for comparison. All the MOFs prepared were tested for CO<sub>2</sub> uptake at 195 K. It was observed that the sorption behavior of MTV-UiO-66-AB was similar to that of UiO-66-B regardless of the ratio between BDC-H and BDC-NH<sub>2</sub>. Interestingly, the CO<sub>2</sub> uptake behavior of MTV-UiO-66-A ⊂ B resembled that of UiO-66-B while the isotherm for MTV-UiO-66-B ⊂ A resembled that of the physical mixture of UiO-66-A and UiO-66-B. In the case of H<sub>2</sub>O uptake measured at 298 K, it was found that the uptake behavior of MTV-MOFs showed a similar trend to that of CO<sub>2</sub>. These pieces of evidence of CO<sub>2</sub> and H<sub>2</sub>O uptake support that the sorption behavior of MTV-UiO-66-AB was strongly affected by the specific functionality present in the outer part of its crystal. These findings were also supported by visualization of iodine sorption using scanning transmission electron microscopy–electron energy loss spectroscopy (STEM–EELS).

In previous work, MTV-MOFs were studied to improve the selectivity and adsorption performance depending on their number, kind, and ratio of functional linkers.<sup>9–11</sup> Herein, this study focuses on the spatial location of functionality in MTV-MOFs and its effect on the unique sorption properties derived therefrom. Other studies have synthesized core/shell-type MOFs using different combinations of functionalities and then showed the properties derived from the structure.<sup>12–15</sup>

However, our study adopts the core/shell type to decipher the unique sorption properties of MTV-MOFs and provides a new way to facilitate the spatially localized heterogeneities in MTV-MOFs, contributing unexpected properties.

## RESULTS AND DISCUSSION

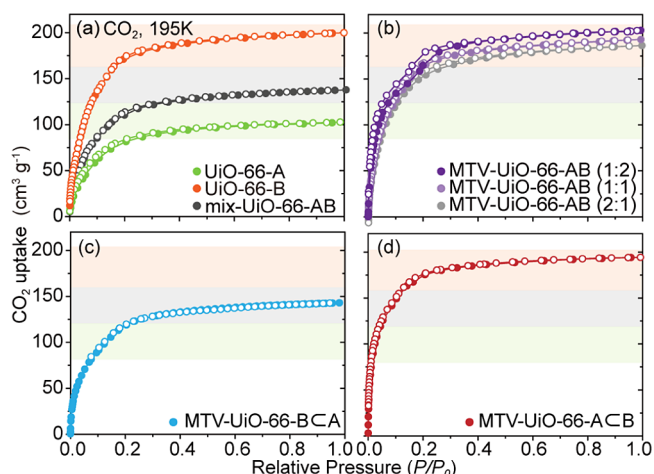
UiO-66 was selected as the basal structure to make MTV-MOFs because it has strong bonding between the organic linkers and the zirconium clusters maintaining its structure after the gas uptake process and even under accelerated electrons in TEM to observe the spatial locality of functionalities. We also choose BDC-NH<sub>2</sub> and BDC-H as the target functional linkers because they have different interactions with polar adsorbents and clearly demonstrate their effect depending on their spatial location in the MTV-MOF. We prepared 6 types of UiO-66 s to decipher the distinct adsorption behavior of MTV-MOF-AB (Scheme S1); three MTV-MOFs (MTV-UiO-66-AB, MTV-UiO-66-A ⊂ B, and MTV-UiO-66-B ⊂ A); and three homogeneous MOFs (UiO-66-A, UiO-66-B, and mix-UiO-66-AB). MTV-UiO-66-A ⊂ B and MTV-UiO-66-B ⊂ A were synthesized by the nanoparticle-mediated nucleation method (Figure S1).<sup>16</sup> UiO-66-A and UiO-66-B were synthesized from BDC-H and BDC-NH<sub>2</sub>, respectively, in DMF containing dissolved ZrCl<sub>4</sub> at 120 °C. Pt nanoparticles (Pt NPs) were subsequently attached on their external surfaces inducing seed growth for outer MOF layers.<sup>17</sup> The outer MOF layers were formed in the synthetic solution having dispersed UiO-66-A and UiO-66-B particles to

give MTV-UiO-66-A  $\subset$  B and MTV-UiO-66-B  $\subset$  A, respectively. When the UiO-66-A expanded to MTV-UiO-66-A  $\subset$  B, the color turned yellow from white, while the yellow color of UiO-66-B faded out as it became MTV-UiO-66-B  $\subset$  A. MTV-UiO-66-AB was prepared by mixing the two linkers of BDC-H and BDC-NH<sub>2</sub> in a preparation solution. The mix-UiO-66-AB was prepared by physically mixing UiO-66-A and UiO-66-B particles. All samples were washed using DMF and MeOH and dried in vacuo overnight. The products were characterized using scanning electron microscopy (SEM), STEM, EELS, powder X-ray diffraction (PXRD), N<sub>2</sub> adsorption, and nuclear magnetic resonance (NMR) spectroscopy. These techniques indicate the morphology, elemental distribution and ratio of functional groups, crystallinity, and porosity of samples. The SEM images showed that all types of MOF samples had a uniform octahedral shape of UiO-66 (Figure S2). The particles of MTV-UiO-66-A  $\subset$  B and MTV-UiO-66-B  $\subset$  A showed 450 and 400 nm diameter which were enlarged from UiO-66-A (300 nm) and UiO-66-B (350 nm), respectively. The EELS mapping showed the positions of the constituents in MTV-MOFs, while the STEM images showed their sizes and morphologies (Figure 1a–i). Zirconium, which constitutes the metal oxide cluster of UiO-66, was enriched throughout the whole crystals. The locations of NH<sub>2</sub>-functional groups in MTV-MOFs were determined by the locations of nitrogen. MTV-UiO-66-AB showed evenly distributed NH<sub>2</sub>-functional groups over the whole crystal, as shown in Figure 1b,c. In contrast, the NH<sub>2</sub>-functional groups were clearly concentrated in the outer layers of MTV-UiO-66-A  $\subset$  B with a thickness of 150 nm (Figure 1e,f). In the case of MTV-UiO-66-B  $\subset$  A, an outer layer of 50 nm thickness having H-functional groups was formed around UiO-66-B (Figure 1h,i). These results confirmed that the locations of the NH<sub>2</sub>-functional groups were partitioned and that they were consistent with the change in the size of the MOF crystals, as shown in the SEM images (Figure S2). As shown in Figures S3–S8, the EELS images indicated the presence or absence of NH<sub>2</sub>-functional groups in homogeneous MOFs. The entire crystal of UiO-66-B consisted of NH<sub>2</sub>-functional groups, which was not the case for UiO-66-A crystals. This was also evident in mix-UiO-66-AB, where UiO-66-A and UiO-66-B were physically mixed. The volumetric contents percentage of the inner and outer parts of the MTV-MOFs were back-calculated, as shown in Table S1. Fourier transform infrared (FT-IR) spectra were also collected to find out the presence of NH<sub>2</sub>-functional groups in MTV-MOFs, UiO-66-B, and mix-UiO-66-AB (Figure S9).

The powder X-ray diffraction (PXRD) patterns of the MTV-MOFs showed sharp diffraction lines that are evidence of high crystallinity, and the coincidence of the diffraction lines matched with simulated UiO-66 patterns that clearly indicate the preservation of the UiO-66 structure regardless of the functional localization in the crystals (Figures 1j and S10). The permanent porosities of MTV-MOFs were confirmed by measuring the N<sub>2</sub> adsorption that showed type-I isotherms similar to those of UiO-66 (Figures 1k and S11). The MTV-MOFs had similar Langmuir surface areas with intermediate values between UiO-66-A and UiO-66-B (Table S2). The digested-<sup>1</sup>H NMR measurements were performed to analyze the presence and ratio of BDC-H and BDC-NH<sub>2</sub> linkers in MTV-MOFs. The samples were prepared by adding strong acid in DMSO-d<sub>6</sub> to disconnect the linkers and metal oxide clusters of MTV-MOFs. Their <sup>1</sup>H NMR spectra are shown in

Figures 1l and S12–S17. The spectra of MTV-UiO-66-AB, MTV-UiO-66-A  $\subset$  B, and MTV-UiO-66-B  $\subset$  A contained peaks at (a) 8.05 ppm and (b–d) 7.8, 7.4, and 7.05 ppm, corresponding to BDC-H and BDC-NH<sub>2</sub>. The integration of these peaks revealed the ratio of BDC-H and BDC-NH<sub>2</sub> in all MOFs as noted in Table S3, which indicates that all MTV-MOFs have a similar ratio of 1:2 for BDC-H and BDC-NH<sub>2</sub>. This value corresponded to the ratio back-counted from the volume fraction of the inner and outer parts of the MTV-MOFs in EELS mapping images (Table S1). Based on the similar ratio of BDC-H and BDC-NH<sub>2</sub> in all the MTV-MOFs, we can solely compare the impacts of spatially localized NH<sub>2</sub>-functional groups in MTV-MOFs on their unique gas sorption properties.

We measured the CO<sub>2</sub> uptake properties to analyze the unusual sorption behavior of MTV-MOFs. CO<sub>2</sub> was selected as the adsorbate because it covalently bonds with the primary amine, which can improve the CO<sub>2</sub> uptake ability in the MTV-MOF system.<sup>18–20</sup> We examined the CO<sub>2</sub> sorption of MTV-MOFs and homogeneous MOFs at 195 K. All the samples were activated by being evacuated at 120 °C for 24 h in a vacuum before measurements. As shown in Figure 2a, UiO-66-



**Figure 2.** CO<sub>2</sub> sorption isotherm of (a) UiO-66-A, UiO-66-B, and mix-UiO-66-AB, (b) MTV-UiO-66-AB (1:2), MTV-UiO-66-AB (1:1), MTV-UiO-66-AB (2:1), (c) MTV-UiO-66-B  $\subset$  A, and (d) MTV-UiO-66-A  $\subset$  B. \*MTV-UiO-66-AB (BDC-H/BDC-NH<sub>2</sub>) were prepared by different ratios of BDC-H and BDC-NH<sub>2</sub> linkers.

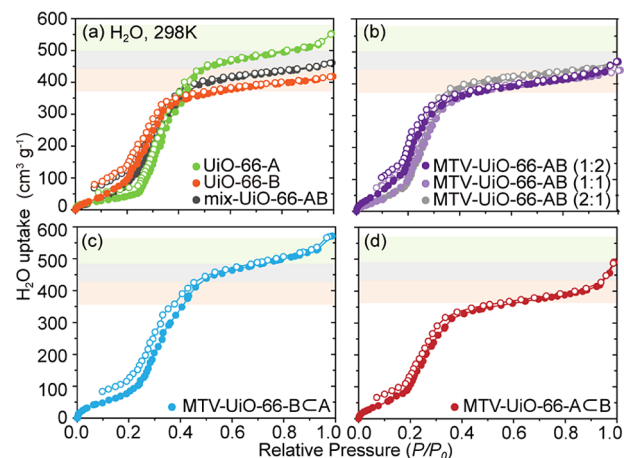
B, which was entirely composed of NH<sub>2</sub>-functional groups, showed higher total uptake than UiO-66-A. The result is due to the strong interaction between the NH<sub>2</sub>-groups and CO<sub>2</sub> and is similar to the results of previous studies.<sup>21–27</sup> In the case of mix-UiO-66-AB, the total uptake was intermediate to those of UiO-66-A and UiO-66-B. Although MTV-UiO-66-AB contained a similar amount of NH<sub>2</sub>-groups as mix-UiO-66-AB, its uptake behavior was similar to that of UiO-66-B. We also prepared multiple MTV-UiO-66-AB samples having different amine ratios, which were denoted as MTV-UiO-66-AB (X:Y), where X and Y are the ratios of BDC-H and BDC-NH<sub>2</sub>, respectively. MTV-UiO-66-AB(1:2), MTV-UiO-66-AB(1:1), MTV-UiO-66-AB(2:1), MTV-UiO-66-AB(9:1), and MTV-UiO-66-AB(33:1) have the BDC-NH<sub>2</sub> content of 68.8, 46.7, 32.6, 9.9, and 2.9%, respectively (Table S3). Interestingly, the MTV-UiO-66-AB (X:Y) samples with a BDC-NH<sub>2</sub> content higher than 2.9% showed a similar uptake behavior and are also

comparable to that of UiO-66-B (Figures 2b and S18). These results indicate that the CO<sub>2</sub> uptake is similar regardless of the ratio when functional groups of more than a certain number exist to facilitate the MTV-system. Therefore, we considered that the spatial location of NH<sub>2</sub>-functional groups might be an important factor for determining the unusual sorption behavior of MTV-MOFs and tested the CO<sub>2</sub> uptake behaviors of MTV-UiO-66-A  $\subset$  B and MTV-UiO-66-B  $\subset$  A. We found that the isotherm of MTV-UiO-66-A  $\subset$  B was similar to that of UiO-66-B and all of the MTV-UiO-66-AB(X:Y) samples regardless of the ratio of NH<sub>2</sub>-functional groups. On the other hand, the isotherm of MTV-UiO-66-B  $\subset$  A was similar to that of the physical mixture of UiO-66-A and UiO-66-B (mix-UiO-66-AB).

Based on the isotherms shown in Figure 2, we classified each of the MOFs tested into two groups: one containing NH<sub>2</sub>-functional groups at the outer part [MTV-UiO-66-A  $\subset$  B, UiO-66-B, and MTV-UiO-66-AB(X:Y)] and the other without NH<sub>2</sub>-functional groups at the outer part (MTV-UiO-66-B  $\subset$  A and UiO-66-A). The MOFs containing NH<sub>2</sub>-functional groups at the outer part of the crystal showed similar CO<sub>2</sub> uptake behaviors whereas other samples showed the uptake behaviors that originated solely from the amount of BDC-H and BDC-NH<sub>2</sub> like the physical mixture (mix-UiO-66-AB). These results indicate that highly interactive functional groups present in the outer part dominantly influence the uptake behavior of the entire crystal, whereas those present in the inner part just contribute as much as its ratio. Considering that highly interactive functional groups spatially exist throughout the MTV-MOF crystals, the unique uptake behavior of MTV-MOF originates from the “outer part” rather than the “inner part”.

The origin of the uniqueness was further investigated by comparing the adsorption behaviors in low relative pressure, which is presented in the logarithm scale in Figure S19. This graph focuses more on monolayer and multilayer adsorption of CO<sub>2</sub> in the samples. The first slope in the initial sorption stage presented a monolayer adsorption behavior and the second slope in the following stage presented a multilayer adsorption behavior in the pores of the samples. The amount of monolayer adsorption in MTV-UiO-66-A  $\subset$  B (48.6 cc g<sup>-1</sup> at  $P/P_0 = 0.004$ ) was much higher than MTV-UiO-66-AB (29.5 cc g<sup>-1</sup> at  $P/P_0 = 0.006$ ), UiO-66-B (29.2 cc g<sup>-1</sup> at  $P/P_0 = 0.004$ ), MTV-UiO-66-B  $\subset$  A (27.0 cc g<sup>-1</sup> at  $P/P_0 = 0.005$ ), UiO-66-A (9.6 cc g<sup>-1</sup> at  $P/P_0 = 0.003$ ), and mix-UiO-66-AB (9.4 cc g<sup>-1</sup> at  $P/P_0 = 0.003$ ). Moreover, the transition pressures from monolayer to multilayer adsorption in MTV-UiO-66-A  $\subset$  B ( $P/P_0 = 0.004$ ) and MTV-UiO-66-AB ( $P/P_0 = 0.007$ ) were much lower than those in MTV-UiO-66-B  $\subset$  A ( $P/P_0 = 0.02$ ), UiO-66-A ( $P/P_0 = 0.04$ ), UiO-66-B ( $P/P_0 = 0.02$ ), and mix-UiO-66-AB ( $P/P_0 = 0.03$ ). We speculate that these behaviors are originated by the sequential interrelation between the pore with a highly affinitive -NH<sub>2</sub> group (-B) and the pore with open space (-A). In the case of MTV-UiO-66-AB and MTV-UiO-66-A  $\subset$  B, the CO<sub>2</sub> molecules traveling into the samples would have high possibility to interact first with the -NH<sub>2</sub> group and then move into the open space. This enables a larger amount of monolayer adsorption and lower pressure transition to multilayer adsorption than other samples including homogeneous MOFs and MTV-UiO-66-B  $\subset$  A. We believe that this is the case of the communication sequence for the guest molecules coming into MTV-MOF samples.

We examined the H<sub>2</sub>O uptake properties to confirm the dominant influence of functional groups in the outer part of the MTV-MOFs. Because the NH<sub>2</sub>-functional groups influence the affinity with H<sub>2</sub>O in MOFs, we tested H<sub>2</sub>O uptake of the MTV-MOFs and homogeneous MOFs at 298 K under relative atmospheric pressure. All the samples were activated by evacuating at 120 °C for 24 h before measurements. Figure 3a

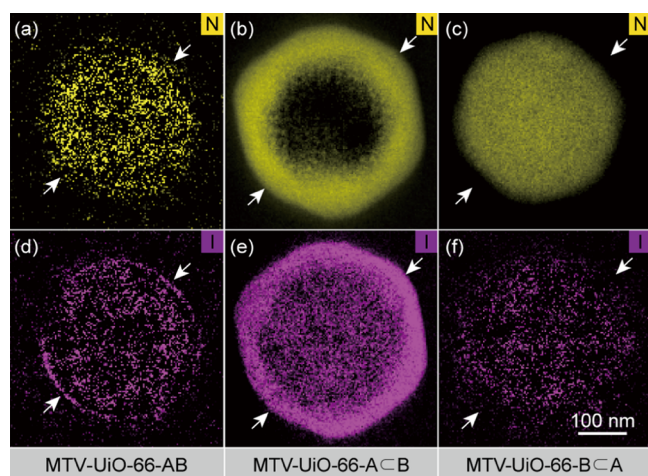


**Figure 3.** H<sub>2</sub>O sorption isotherm of (a) UiO-66-A, UiO-66-B, and mix-UiO-66-AB, (b) MTV-UiO-66-AB (1:2), MTV-UiO-66-AB (1:1), MTV-UiO-66-AB (2:1), (c) MTV-UiO-66-B  $\subset$  A, and (d) MTV-UiO-66-A  $\subset$  B. \*MTV-UiO-66-AB (BDC-H/BDC-NH<sub>2</sub>) were prepared by different ratios of BDC-H and BDC-NH<sub>2</sub> linkers.

shows a comparison of the isotherms for homogeneous MOFs. In contrast to the CO<sub>2</sub> isotherms at 195 K, UiO-66-B (411.3 cm<sup>3</sup>g<sup>-1</sup>) showed lower total uptake than UiO-66-A (517.7 cm<sup>3</sup>g<sup>-1</sup>) at  $P/P_0 = 0.95$ . This is because the presence of NH<sub>2</sub>-groups make the pore smaller, decreasing the total water uptake rather than increasing affinity at room temperature.<sup>28–31</sup> However, the increased affinity makes H<sub>2</sub>O uptake start at a lower relative pressure than UiO-66-A (Figure 3a). Figure 3b shows that MTV-MOF-AB(X:Y) had a similar total uptake in all amine ratios and their isotherms were similar to those of MOF-B, which is consistent with the CO<sub>2</sub> isotherm results (Figure 2b). MTV-UiO-66-B  $\subset$  A and MTV-UiO-66-A  $\subset$  B were tested to investigate the spatial effect of the NH<sub>2</sub>-functional groups in or out of MTV-MOF particles on the H<sub>2</sub>O adsorption capacity (Figure 3c,d) as investigated in CO<sub>2</sub> uptake analysis (Figure 2c,d). The total uptake and adsorption isotherm of MTV-UiO-66-B  $\subset$  A resembled those of UiO-66-A (Figure 3c). Similarly, MTV-UiO-66-A  $\subset$  B showed a correlative isotherm to those of MTV-UiO-66-AB(X:Y) and UiO-66-B (Figure 3d). This indicates that the uptake behavior of the outer part overwhelms that of the whole crystal of MTV-MOF.

The effect of polarity on these usual uptake behaviors of MTV-MOFs was investigated by comparing the isotherms of CO<sub>2</sub> and N<sub>2</sub> at their boiling temperature (Figures 2 and S11) and the isotherms of H<sub>2</sub>O and CO<sub>2</sub> at room temperature (298 K) (Figures 3 and S20). These comparisons showed that the unusual behaviors of MTV-MOFs were observed when the adsorbate had sufficient interaction with functional groups at a given temperature. Therefore, the results make clear that spatial heterogeneity combined with the intermolecular interaction with functional groups highly effect on the unusual uptake behaviors of MTV-MOFs.

The finding, that the uptake in the outer part affects the inner part in MTV-MOFs, was intuitively demonstrated by using the iodine sorption method. By exposing MTV-MOF particles to iodine, its vapor diffused into the MTV-MOFs can be visualized by elemental mapping images obtained in STEM–EELS measurements. As iodine has a high atomic number of 53, its presence is readily detected by STEM measurements.<sup>32–35</sup> Samples were prepared by placing each MOF sample on a separate TEM grid followed by being exposed to iodine for 2 days at room temperature in sealed vials to sufficiently diffuse the iodine vapor into the MOF crystals. The position of BDC-NH<sub>2</sub> was indicated by the yellow color in the nitrogen mapping images while the boundaries of the entire crystal are indicated by arrows in Figure 4a–c. The



**Figure 4.** EELS elemental mapping images of nitrogen (top) and iodine (bottom) for spatially localized MTV-MOFs. (a,d) MTV-UiO-66-AB, (b,e) MTV-UiO-66-A C B, and (c,f) MTV-UiO-66-B C A.

iodine mapping images in Figure 4d–f indicate the presence of iodine by purple color. Because the NH<sub>2</sub>-functional groups have polar hydrophilic interactions, iodine tends to be observed intensively in the NH<sub>2</sub>-rich areas, as shown in Figure 4d–f.<sup>36–39</sup> In the case of MTV-UiO-66-AB, iodine was diffused throughout the crystal according to the distribution of the NH<sub>2</sub>-functional groups (Figure 4d). Interestingly, MTV-UiO-66-A C B exhibited highly intensive iodine signals in the inner and outer parts of the crystal (Figure 4e). Because of the presence of the NH<sub>2</sub>-functional groups in the outer part, the purple coloration of the outer part was more intense than that of the inner part. In contrast, iodine adsorbed by MTV-UiO-66-B C A majorly remains in the inner part, not in the outer part (Figure 4f). It should be noted that the intercomparison between different images for different samples is not allowed as each image has different brightness and contrast. Comparing the relative intensity of purple coloration in the pore with open space (-A) versus that in the pore with -NH<sub>2</sub> group (-B) in each sample, MTV-UiO-66-A C B has a higher intensity than MTV-UiO-66-B C A (Figure 4e,f). These results visualize that the uptake behavior of the outer part with the -NH<sub>2</sub> group (-B) affects that of the inner part with open space (-A) of the crystal in MTV-MOF. Therefore, we believe that the unique behavior of MTV-UiO-66-AB is influenced more strongly by the presence of the NH<sub>2</sub>-functional groups in the “outer part” rather than the “inner part” of the crystal.

## CONCLUSIONS

In this study, the unique adsorption properties of MTV-MOF were analyzed by implementing the spatial localization of functional groups in MTV-UiO-66-AB. Specifically, the locality was investigated by synthesizing MTV-UiO-66-A C B and MTV-UiO-66-B C A crystals having NH<sub>2</sub>-functionalities in the outer and inner parts of the crystals, respectively. We tested the sorption properties of MTV-UiO-66-AB, MTV-UiO-66-A C B, and MTV-UiO-66-B C A using two adsorbates (CO<sub>2</sub> and H<sub>2</sub>O) and eventually found that the uptake behavior in the outer part strongly influences that of the whole crystal in MTV-MOFs. Furthermore, this finding was also visualized using iodine adsorption and STEM-EELS observation. Considering these results, we propose to focus on the localization of functional heterogeneity in MTV-MOFs to explore their unique uptake behaviors.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/jacs.2c12207>.

Detailed experimental and experimental methods; additional SEM images; STEM-EELS images; FT-IR spectra; PXRD; N<sub>2</sub> adsorption; and NMR spectroscopy (PDF)

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#S.K. and U.R. contributed equally to this paper.

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## Notes

The authors declare no competing financial interest.

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